

Potassium sodium phyllosilicate,
 $K_{1.33}Na_{0.67}Si_2O_5$ Anthony L. Spek^{a*} and
Bernard H. W. S. de Jong^b^aFaculty of Sciences, Bijvoet Center for Biomolecular Research, Department of Crystal and Structural Chemistry, Utrecht University, Padualaan 8, 3584 CH Utrecht, The Netherlands, and ^bFaculty of Geosciences, Utrecht University, Budapestlaan 4, 3584 CD, Utrecht, The Netherlands

Correspondence e-mail: a.l.spek@chem.uu.nl

Key indicators

Single-crystal X-ray study
 $T = 150\text{ K}$
Mean $\sigma(\text{Si}-\text{O}) = 0.004\text{ \AA}$
 R factor = 0.043
 wR factor = 0.118
Data-to-parameter ratio = 11.4For details of how these key indicators were automatically derived from the article, see <http://journals.iucr.org/e>.

The two-dimensional network topology of the mixed alkali title compound, $K_{1.33}Na_{0.67}Si_2O_5$, is found to differ from that of the end-member compositions, $Na_2Si_2O_5$ and $K_2Si_2O_5$. The network of the silicate sheet consists of rings with four, six and eight SiO_4 tetrahedra. The cations are sandwiched between the silicate sheets.

Comment

In previous papers, we have described the preparation and structures of a number of crystalline alkali and mixed alkali disilicates (phyllosilicates) with chemical formula $R_2O \cdot 2SiO_2$ ($R = \text{Li, Na, K, Rb}$ and Cs ; de Jong *et al.*, 1994, 1996, 1998, 2000; Veldman *et al.*, 1995). Not all combinations of alkali metals form mixed alkali disilicates at ambient pressure. The impossible combinations are (Li,Na), (K,Rb), (K,Cs) and (Rb,Cs) disilicates. Their cationic field strength difference is too small according to a theory proposed by Dietzel (1983).

There are many reasons why the study of these materials is useful. Our reason was to find an explanation for the variation in hygroscopicity of mixed alkali disilicate glasses. These glasses may be used as model compounds for dried soluble silicates (Weldes & Lange, 1969), for which there was an industrial need to make them non-hygroscopic. Though

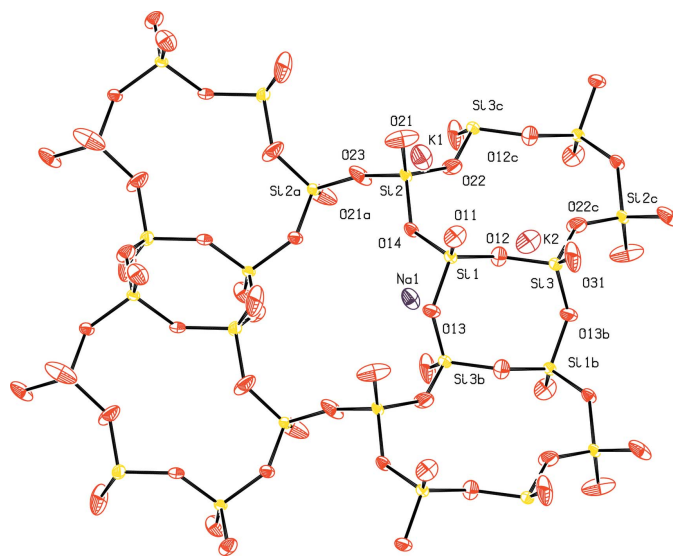
Received 1 July 2005
Accepted 12 July 2005
Online 27 July 2005

Figure 1
View of part of the Si_2O_5 two-dimensional network consisting of four-, six- and eight-membered rings of SiO_4 tetrahedra and crystallographically independent cations, along with the atom-numbering scheme. Displacement ellipsoids are drawn at the 70% probability level. [Symmetry codes: (a) $-x, y, -\frac{1}{2} - z$; (b) $\frac{1}{2} - x, \frac{1}{2} - y, -z$; (c) $\frac{1}{2} - x, \frac{3}{2} - y, -z$.]

soluble silicates tend to have a lower alkali-to-silica ratio, we chose as a homologous the alkali disilicates. The advantage of this system is that it can form, next to glasses, crystalline disilicates, which have hygroscopicities similar to their glassy congeners. Such crystalline materials can of course be analyzed structurally as well as chemically in much more detail to discover the causes for the observed very large variations in hygroscopicity. Unfortunately, no obvious structure–hygroscopy relation was found.

From a crystal chemical perspective, these single and mixed alkali systems turned out to be surprisingly rich in network topologies. Though compositionally constrained to structures in which each tetrahedron contains three O atoms connected to two Si atoms, and one O atom connected to only one Si atom, they formed not only sheets but also three-dimensional networks and one-dimensional ribbons. One of the few remaining phases we synthesized in this system was the mixed (K,Na) disilicate structure, which has recently received renewed attention (Rakic & Kahlenberg, 2001; Rakic *et al.*, 2003*a,b*).

The title compound, $K_{1.33}Na_{0.67}Si_2O_5$, is a sheet silicate (Figs. 1 and 2) with four-, six- and eight-membered rings of SiO_4 tetrahedra. Na is five-coordinated and K six-coordinated by oxygen.

Experimental

The crystalline phase can be made without any special precaution by mixing K_2CO_3 (1 mol), Na_2CO_3 (1 mol) and SiO_2 (4 mol), calcining the mixture at 800 K to remove the CO_2 , and melting the resulting $K_2O \cdot Na_2O \cdot 4SiO_2$ (m.p. 878 K; Kracek, 1932) mixture at 1000 K, followed by cooling at a rate of 0.5 K min^{-1} . Samples were immersed in petroleum jelly to prevent absorption of moisture.

Crystal data

$K_{1.33}Na_{0.67}Si_2O_5$
 $M_r = 203.64$
 Monoclinic, $C2/c$
 $a = 19.6816$ (15) Å
 $b = 7.2656$ (6) Å
 $c = 12.5710$ (15) Å
 $\beta = 117.82$ (1)°
 $V = 1589.9$ (3) Å³
 $Z = 12$

$D_x = 2.552 \text{ Mg m}^{-3}$
 Mo $K\alpha$ radiation
 Cell parameters from 9 reflections
 $\theta = 10\text{--}14^\circ$
 $\mu = 1.71 \text{ mm}^{-1}$
 $T = 150 \text{ K}$
 Block, colorless
 $0.3 \times 0.3 \times 0.2 \text{ mm}$

Data collection

Enraf–Nonius TurboCAD-4 diffractometer
 ω scans
 Absorption correction: none
 2841 measured reflections
 1406 independent reflections
 1094 reflections with $I > 2\sigma(I)$
 $R_{int} = 0.068$

$\theta_{max} = 25^\circ$
 $h = -23 \rightarrow 8$
 $k = -8 \rightarrow 8$
 $l = -13 \rightarrow 14$
 1 standard reflections
 frequency: 60 min
 intensity decay: none

Refinement

Refinement on F^2
 $R[F^2 > 2\sigma(F^2)] = 0.043$
 $wR(F^2) = 0.118$
 $S = 1.09$
 1406 reflections
 123 parameters

$w = 1/[\sigma^2(F_o^2) + (0.0465P)^2 + 11.2001P]$
 where $P = (F_o^2 + 2F_c^2)/3$
 $(\Delta/\sigma)_{max} = 0.001$
 $\Delta\rho_{max} = 0.65 \text{ e \AA}^{-3}$
 $\Delta\rho_{min} = -0.92 \text{ e \AA}^{-3}$

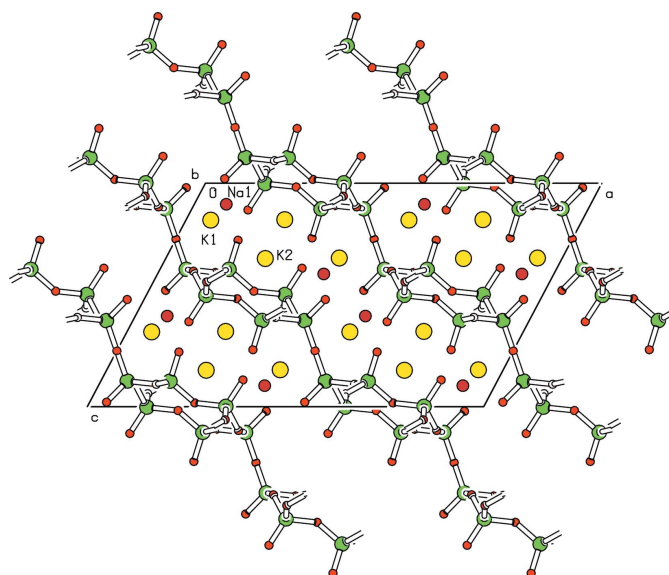


Figure 2
 Projection of the structure down the b axis, illustrating the layer structure with the cations sandwiched in between.

Table 1

Selected interatomic distances (Å).

| | | | |
|-----------------------|-------------|------------------------|-----------|
| Si1—O11 | 1.559 (4) | K1—O14 ^{iv} | 2.904 (4) |
| Si1—O12 | 1.636 (5) | K1—O13 ^{iv} | 3.353 (5) |
| Si1—O13 | 1.634 (5) | K1—O31 ^v | 2.637 (5) |
| Si1—O14 | 1.650 (4) | K2—O11 | 2.614 (4) |
| Si2—O14 | 1.647 (4) | K2—O31 | 2.607 (5) |
| Si2—O21 | 1.550 (6) | K2—O12 ^{iv} | 2.884 (4) |
| Si2—O22 | 1.620 (5) | K2—O22 ^{vi} | 2.850 (5) |
| Si2—O23 | 1.6024 (16) | K2—O22 ^{vi} | 2.980 (6) |
| Si3—O12 | 1.643 (4) | K2—O12 ^{vi} | 3.127 (4) |
| Si3—O31 | 1.550 (4) | Na1—O11 | 2.495 (5) |
| Si3—O13 ⁱ | 1.630 (5) | Na1—O13 | 2.702 (5) |
| Si3—O22 ⁱⁱ | 1.638 (5) | Na1—O21 ^{vii} | 2.356 (5) |
| K1—O11 | 2.563 (5) | Na1—O21 ⁱⁱⁱ | 2.271 (7) |
| K1—O21 | 2.562 (5) | Na1—O31 ^{vi} | 2.291 (5) |
| K1—O14 ⁱⁱⁱ | 2.899 (5) | | |

Symmetry codes: (i) $-x + \frac{1}{2}, -y + \frac{1}{2}, -z$; (ii) $-x + \frac{1}{2}, -y + \frac{3}{2}, -z$; (iii) $-x, -y + 1, -z$; (iv) $x, -y + 1, +z + \frac{1}{2}$; (v) $-x + \frac{1}{2}, +y + \frac{1}{2}, -z + \frac{1}{2}$; (vi) $-x + \frac{1}{2}, +y - \frac{1}{2}, -z + \frac{1}{2}$; (vii) $x, y - 1, z$.

The displacement parameters of O23 were restrained to be approximately isotropic.

Data collection: locally modified *CAD-4 Software* (Enraf–Nonius, 1989); cell refinement: *SET4* (de Boer & Duisenberg, 1984); data reduction: *HELENA* (Spek, 1997); program(s) used to solve structure: *DIRDIF99* (Beurskens *et al.*, 1999); program(s) used to refine structure: *SHELXL97* (Sheldrick, 1997); molecular graphics: *PLATON* (Spek, 2003); software used to prepare material for publication: *PLATON*.

This work was supported in part (ALS) by the Council for the Chemical Sciences of the Netherlands Organization for Scientific Research (CW-NWO).

References

- Beurskens, P. T., Beurskens, G., de Gelder, R., García-Granda, S., Gould, R. O., Israel, R. & Smits, J. M. M. (1999). *The DIRDIF99 Program System*. Technical Report of the Crystallography Laboratory, University of Nijmegen, The Netherlands.

- Boer, J. L. de & Duisenberg, A. J. M. (1984). *Acta Cryst.* **A40**, C-410.
- Dietzel, A. (1983). *Phys. Chem. Glasses*, **24**, 172–180.
- Enraf–Nonius (1989). *CAD-4 Software*. Version 5. Enraf–Nonius, Delft, The Netherlands.
- Jong, B. H. W. S. de, Slaats, P. G. G., Super, H. T. J., Veldman, N. & Spek, A. L. (1994). *J. Non-Cryst. Solids*, **176**, 64–71.
- Jong, B. H. W. S. de, Super, H. T. J., Frijhoff, R. M., Spek, A. L. & Nachtegaal, G. (2000). *Z. Kristallogr.* **215**, 397–405.
- Jong, B. H. W. S. de, Super, H. T. J., Spek, A. L., Veldman, N., Nachtegaal, G. & Fischer, J. C. (1998). *Acta Cryst.* **B54**, 568–577.
- Jong, B. H. W. S. de, Super, H. T. J., Spek, A. L., Veldman, N., van Wezel, W. & van der Mee, V. (1996). *Acta Cryst.* **B52**, 770–776.
- Kracek, F. C. (1932). *J. Phys. Chem.* **36**, 2529–2542.
- Rakic, S. & Kahlenberg, V. (2001). *Solid State Sci.* **5**, 659–667.
- Rakic, S., Kahlenberg, V. & Schmidt, B. C. (2003a). *Solid State Sci.* **5**, 473–480.
- Rakic, S., Kahlenberg, V. & Schmidt, B. C. (2003b). *Z. Kristallogr.* **218**, 413–420.
- Sheldrick, G. M. (1997). *SHELXL97*. University of Göttingen, Germany.
- Spek, A. L. (1997). *HELENA*. Utrecht University, The Netherlands.
- Spek, A. L. (2003). *J. Appl. Cryst.* **36**, 7–13.
- Veldman, N., Spek, A. L., Super, H. T. J. & de Jong, B. H. W. S. (1995). *Acta Cryst.* **C51**, 1972–1975.
- Weldes, H. K. & Lange, K. R. (1969). *Ind. Eng. Chem.* **61**, 4, 29–44.