

The influence of free-carrier concentration on the PEC etching of GaN: A calibration with Raman spectroscopy

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Abstract

Large areas of electrically active regions of in-homogeneities have been revealed by the electroless photo-etching (PEC) method in GaN layers grown by the hydride vapor phase epitaxy (HVPE) technique.

Variations in the local etch rate have been correlated with the variations in the free-carrier concentrations as determined by micro-Raman spectroscopy. The etch rate decreased linearly with the log of the carrier concentration. The latter could change by more than two orders of magnitude on the same sample.

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1. Introduction

Growth of homo-epitaxial layers of GaN by hydride vapor phase epitaxy (HVPE) is an attractive solution for obtaining large-area crystals of high structural, electronic and optical quality. However, such layers are not completely free of crystallographic defects and other types of non-homogeneities. Electroless photo-etching (PEC) can reveal not only the crystallographic defects (e.g. dislocations), but also electrically active chemical non-homogeneities (differences in free-carrier concentrations) [1,2]. An advantage of this method is the possibility for either large area or local etching (only the illuminated part of the sample is etched) to map the non-homogeneities on the chosen surface of the sample [3]. However, this method must be calibrated to determine a free-carrier concentra-

tion. In a previous work we found qualitative relationships between the PEC etch rate and Raman spectroscopy-derived free-carriers concentrations [2], higher etching depths corresponding to lower free-carriers concentrations. In this paper we report on a quantitative relationship between these two parameters. Such a relationship has been previously established for GaAs etched using DSL (diluted Sirtl-like) photoetching [4,5]. In those works, the authors have shown that the etch rate is proportional to the logarithm of the free-carrier concentration determined using electron beam induced current (EBIC) method. In this communication we used Raman spectroscopy as a tool to determine directly free-carrier concentrations.

2. Experimental procedure

Thick HVPE homo-epitaxial GaN layers were grown in a home-made, horizontal HVPE system using a quartz reactor, N₂ as the carrier gas and NH₃ as the nitrogen source [6]. Typically the epitaxial layers were grown on

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single-crystalline GaN substrates in [0001] direction. However, at the edge of the substrates the macroscopic growth directions are $\langle 1\bar{1}00 \rangle$ and $\langle 11\bar{2}0 \rangle$. This results in formation of a “frame” on the (0001) surface (which corresponds to so-called wings, as recognized on the (10 $\bar{1}$ 0) longitudinal cross-sections in Ref. [2]) with different morphologies and structural and electrical properties.

To eliminate morphological non-homogeneities in the as-grown HVPE layer (large hillocks are typical for the thick layers) the samples were mechanically polished and subsequently subjected to reactive ion etching (RIE) to remove work damage. Wax spots were placed on the sample surface for the purpose of the etch rate measurements. Photo-etching in stirred aqueous KOH solution was performed using illumination from a 450 W Xe lamp [7,8]. After photo-etching of HVPE-grown GaN, apart from the formation of whisker-like features on dislocations, a nano-size “grass” is frequently formed. The nature of the grass has not been established so far despite the attempts made using TEM on the photo-etched samples [9]. To exclude a possible influence of this etching artifact on the free-carrier concentration and depth measurements, we performed our experiments twice. First, the etch depth and Raman measurements were done along lines chosen on the surface of the sample directly after photo-etching. Then the samples were subjected to a newly developed etching procedure (etching in 1 M KOH solution at 85 °C for 30 s [10]) to remove the protruding features from the surface. Finally the surface profile and Raman measurements were repeated.

The etching depth profiles were recorded using an Alfa-step profiler. Micro-Raman measurements were performed along the same lines with a Jobin–Yvon T64000 spectrometer at ambient conditions using the 514.532 nm line of an argon laser as an excitation source. Spectra were taken from the (0001) Ga surfaces in backscattering configuration. The Raman spot was of the order of 1 μm in diameter. Surface examination was performed using differential interference contrast (DIC) optical microscopy.

3. Results and discussion

3.1. Determination of the free-carrier concentration using Raman spectroscopy

In polar crystals, the long-range coulombian interaction splits polar modes (A_1 and E_1 in the case of GaN) into transversal (TO) and (LO) modes. The LO modes can couple with collective vibrations of free carriers–plasmons resulting in modes called longitudinal plasmon–phonon (LPP) modes. From the position of these modes and knowing the position of the uncoupled LO mode, one can find a plasmon frequency ω_p from the formula (1)

$$\omega_{\pm}^2 = \frac{(\omega_L^2 + \omega_p^2 - \gamma\Gamma) \pm \sqrt{(\omega_L^2 + \omega_p^2 - \gamma\Gamma)^2 - 4\omega_p^2\omega_T^2}}{2}, \quad (1)$$

where ω_{\pm} is the frequency of the high-frequency LPP (LPP+) mode and the low-frequency LPP (LPP–) mode, respectively; ω_L and ω_T are the LO and TO phonon frequencies, respectively; γ is the phonon damping constant and Γ is the electron damping constant. The free-carrier concentration n is directly related to the plasmon frequency ω_p obtained from the LPP modes positions (Eq. (1))

$$n = \frac{\omega_p^2 \varepsilon_{\infty} \varepsilon_0 m^*}{e^2}, \quad (2)$$

where ε_0 is the vacuum permittivity, m^* the effective mass of the electron and e is the electron charge. Because we performed our experiments on the surface perpendicular to the c -axis of the crystal, only the A_1 polar mode was observed. In our calculations we used the following parameters: $\omega(A_1(\text{TO})) = \omega_T = 532 \text{ cm}^{-1}$, $\omega(A_1(\text{LO})) = \omega_L = 736 \text{ cm}^{-1}$, $\varepsilon_{\infty} = 5.35$ and $m^* = 0.2$ of electron mass [11]. We have neglected damping constants. The uncertainty on the free-carrier concentration values determined was between 5% (for the higher values) and 20% (lower values). In this work, the minimum value of the free-carrier concentration which could be determined with a reasonable precision was 10^{17} cm^{-3} . Examples of the Raman spectra of n-GaN are shown in Fig. 1.

3.2. Etching depth and free-carrier concentration profiles

Fig. 2 shows a DIC optical image of the GaN surface of one of the investigated samples after photo-etching with the lines 1–1' and 2–2' along which the etch depth surface profiles were recorded. Fig. 3 shows a comparison between the etching profile along line 2–2' and the free-carrier concentration distribution calculated from the $A_1(\text{LO})$ mode frequency. The profiles of the free-carrier concentration bear a very strong resemblance to the respective profiles of the etch depth (similar results were obtained on three different samples and several linescans), confirming a previously observed qualitative correlation between these values [2]. Subsequently, the etch rates were calculated and the relationship with the corresponding values of the free-carrier concentrations was established. For the investigated range (1×10^{17} – $5 \times 10^{18} \text{ cm}^{-3}$) a linear relationship was found between the logarithm of the free-carrier concentration n (in cm^{-3}) and the etch rate R (in nm/min). The results are presented in Fig. 4. A linear function was fit to the experimental results for each profile. This relationship is of the form

$$R = \left(-\frac{1}{\beta}\right) \ln\left(\frac{n}{n_{\text{th}}}\right), \quad (3)$$

where R is the etch rate, n is the free-carrier concentration, n_{th} is the free-carrier concentration corresponding to $R = 0$ (it is specific to a given sample and can be obtained by extrapolation of the $R = f(n)$ lines) and β is a slope coefficient. As shown in Fig. 4, the slopes of the $R = f(n)$ lines are very similar within the limits of experimental

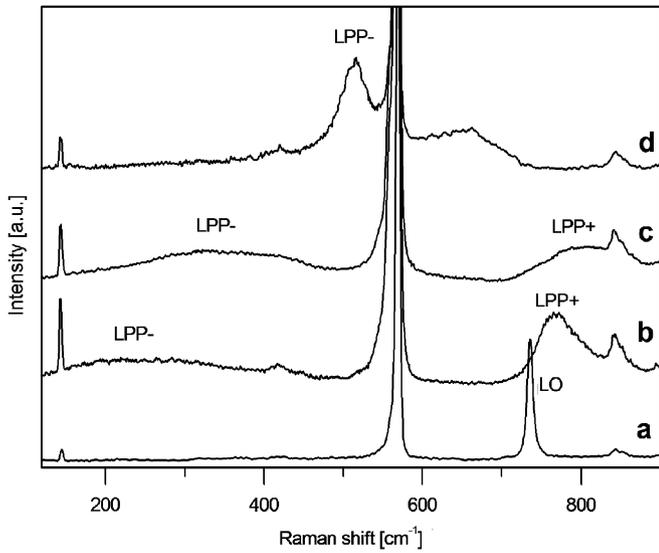


Fig. 1. Raman spectra of n-GaN with different doping levels; (a) LO mode is not shifted and no LPP modes are observed, it means that there is no phonon–plasmon coupling, because the free-carrier concentration is too low and can be estimated as $<10^{17} \text{ cm}^{-3}$; (b) both LPP+ (at 770 cm^{-1}) and LPP- (at 220 cm^{-1}) modes are observed, calculated free-carrier concentration is $1.25 \times 10^{18} \pm 0.05 \text{ cm}^{-3}$; (c) LPP+ (at 803 cm^{-1}) and LPP- (at 326 cm^{-1}) are observed, free-carrier concentration is $2.9 \pm 0.1 \times 10^{18} \text{ cm}^{-3}$; (d) only LPP- mode (at 515 cm^{-1}) is shown, LPP+ was found at 1887 cm^{-1} , calculated free-carrier concentration is $4 \times 10^{19} \text{ cm}^{-3}$.

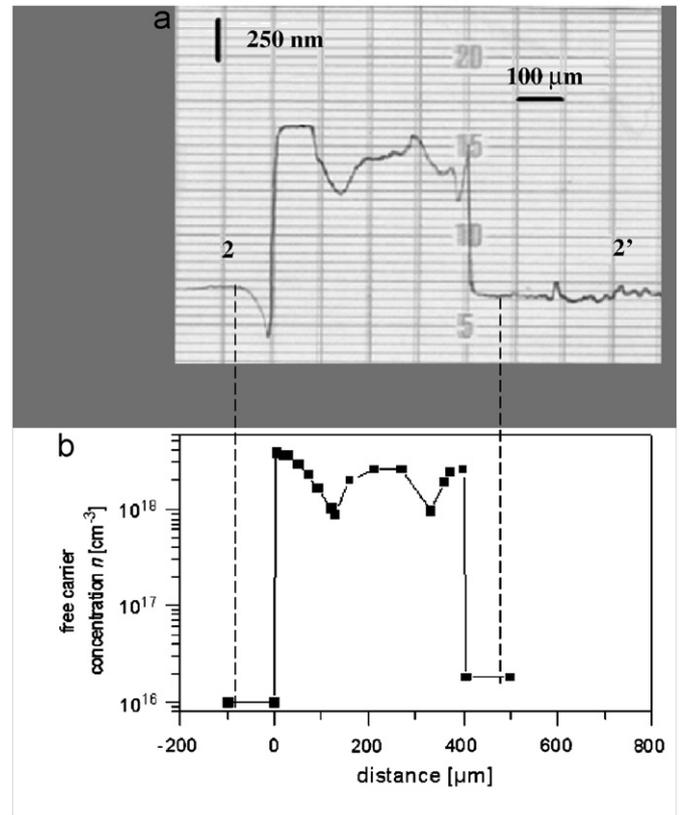


Fig. 3. (a) Comparison of the etch depth profile along the line 2–2' (Fig. 2) performed using a profilometer and (b) free-carrier concentration profile calculated from the position of the LPP+ modes.

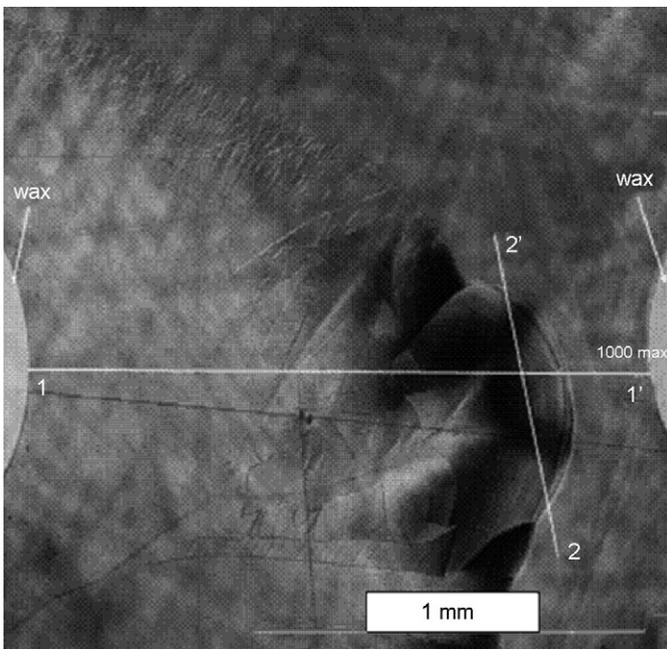


Fig. 2. DIC optical image of the GaN surface after photo-etching with the lines 1–1' and 2–2' along which the etch depth surface profiles were recorded.

error. Lines from profiles 1–1' and 2–2' of sample 1 overlap as would be expected for measurements performed on the same sample (the same n_{th}) but the results from sample 2

are shifted (n_{th} values are different for samples grown separately under different conditions, e.g. $(n_{\text{th}})_{1-1'} = (n_{\text{th}})_{2-2'} = 6 \times 10^{18} \text{ cm}^{-3}$ and $(n_{\text{th}})_{3-3'} = 7 \times 10^{17} \text{ cm}^{-3}$ in Fig. 4). Sample regions with a carrier concentration higher than n_{th} will not be etched. Indeed, on all three samples there were parts of the surface showing no noticeable etching depth (e.g. the “frames” and also some areas in the central part of the samples). The carrier concentrations of these areas are represented by the encircled points lying on the abscissa in Fig. 4 and clearly they do not comply with the linear relationship $R = f(n)$. The explanation of these seemingly puzzling experimental results can be given on the basis of the principles of electroless etching established for GaAs [12] and GaN [13]. Fig. 5 shows a schematic representation of the photocurrent–potential curves for a sample containing areas of different carrier concentrations in a broad range ($n-n^{++}$). The onset of the photocurrent j shifts to higher potential V as the carrier concentration increases ($n \rightarrow n^{++}$) [13]. The rest (or mixed) potential of the system, determined by oxygen reduction at the counter electrode [13], is indicated by V_m . From Fig. 5 it is clear that at this potential the photocurrent, and thus the photoetch rate, decreases as n increases ($j_m^n > j_m^{n++}$) while above n_{th} no etching is observed. The value of n_{th} clearly depends on the particular sample.

